

# Wednesday Afternoon, November 8, 2023

## Applied Surface Science Division

### Room B117-119 - Session AS+CA+EL+EM+SE+SS+TF-WeA

#### Quantitative Surface Analysis I

**Moderators:** David Cant, National Physical Laboratory, UK, Peter Cumpson, University of New South Wales, Christopher Moffitt, Kratos Analytical Inc, Lev Gelb, University of Texas at Dallas

2:20pm **AS+CA+EL+EM+SE+SS+TF-WeA-1 Status of Efforts to Upgrade the Quality of Surface Analysis Data in the Literature**, Donald Baer, Pacific Northwest National Laboratory

Multiple efforts are being undertaken to address a growing presence of faulty surface analysis data and/or analyses appearing in the literature. Issues include bad data, incorrect analysis, and highly incomplete reporting of instrument and analysis parameters. This talk describes the status of four efforts to address some of the issues. Recognition of this problem within the surface analysis community has increased with an understanding that both inexperienced users and increased use of surface analysis methods outside the surface analysis community contribute to the problem. The current efforts build upon decades of development and efforts by standards committees, excellent books and journal publications, websites, short courses, and other efforts. A collection of guides, protocols and tutorials addressing reproducibility issues with a significant focus on XPS was published in JVSTA in 2020/21. A second collection, *Reproducibility Challenges and Solutions II*, with a more general focus on Surface and Interface Analysis was initiated in 2022 and is being finalized. The second collection addresses several techniques, including SIMS, SPM, and UPS, and includes topics such as theoretical modeling and machine learning in data analysis. A second effort focuses on a part of the community less interested in general understanding but needs to answer specific surface analysis questions. A new type of paper called Notes and Insights is being published in the journal Surface and Interface Analysis with the objective to provide incremental bits of useful information of importance to non-expert analysts. Two additional activities are underway to assist with reporting issues. Examination of papers in multiple journals found that instrument and analysis related information needed to assess or reproduce data is often incomplete or absent. To assist authors in reporting instrument parameters, papers describing in some detail related families of commercial instruments are being prepared for Surface Science Spectra. These papers describe the instrument, major components, geometry and provide example data related to common data collection modes. Authors will be able to reference these papers and identify specific modes of instrument operation used in their research. Another parameter reporting activity concerns sample handling before analysis. ISO Technical Committee 201 is developing a series of documents (ISO 20579 parts 1 to 4) on what needs to be reported regarding sample handling, storage, processing, and mounting for analysis. These standards describe what needs to be reported and contain informative annexes that provide information regarding the needs and challenges to proper sample handling to produce reliable surface analysis data.

2:40pm **AS+CA+EL+EM+SE+SS+TF-WeA-2 The behavior of the Shirley background of the Ti 2p spectra across the Ti 1s edge**, Dulce Maria Guzman Bucio, CINVESTAV-Unidad Queretaro, Mexico; D. Cabrera German, Universidad de Sonora, Mexico; O. Cortazar Martinez, J. Raboño Borbolla, CINVESTAV-Unidad Queretaro, Mexico; M. Vazquez Lepe, Universidad de Guadalajara, Mexico; C. Weiland, J. Woicik, National Institute of Standards and Technology; A. Herrera Gomez, CINVESTAV-Unidad Queretaro, Mexico

A wide variety of photoemission spectra display a step-shaped background, called the Shirley-type background, which should be accounted for in the total background signal for reliably assessing chemical composition. However, it cannot be modeled with any method based on extrinsic processes like the inelastic dispersion of the photoelectrons (e.g., Tougaard-type backgrounds). Although its physical origin is still unknown, experimental data suggest that the Shirley-type background is due to phenomena occurring inside the atom [1,2]. To gain insights into those phenomena, we studied the behavior of the Shirley-type background for the Ti 2p photoemission spectra.

In this work, Ti 2p photoemission spectra were acquired with Synchrotron light (at Beamline 7-2 at the Brookhaven National Laboratory) from a clean metallic titanium film (sputtered on a Si (100) substrate) capped with an

ultra-thin aluminum layer. The spectra were collected with 44 excitation energies around the Ti 1s edge. By simultaneously fitting Ti 2p photoemission spectra obtained with excitation energies higher than the Ti 1s edge, we robustly determined the peak structure of the Ti 2p spectra. Outstandingly, the parameter of the Shirley-type background associated with the Ti 2p peak structure is modulated as the photon energy crosses the Ti 1s edge. The relation of this phenomenon with the physical origin of the Shirley background will be discussed. The KVL<sub>2,3</sub> Auger peaks—which overlap with the Ti 2p peaks—do not have a step-shaped background for most of the excitation energies.

Acknowledgments:

Use of the Brookhaven National Laboratory is supported by the U.S. Department of Energy's (DOE) Office of Science.

This work was partially financed by CONACyT Project Fronteras 58518, Mexico.

References:

- [1] A. Herrera-Gomez, D. Cabrera-German, A. Dutoi, M. Vazquez-Lepe, S. Aguirre-Tostado, P. Pianetta, D. Nordlund, O. Cortazar-Martinez, A. Torres-Ochoa, O. Ceballos-Sanchez, L. Gomez-Muñoz, Intensity modulation of the Shirley background of the Cr 3p spectra with photon energies around the Cr 2p edge, Surf. Interface Anal. 50 (2018) 246–252. <https://doi.org/10.1002/sia.6364>.
- [2] A. Herrera-Gomez, Interchannel Coupling with Valence Band Losses as the physical origin of the Shirley background in photoemission spectra (Old title: The unresolved physical origin of the Shirley background in photoemission spectra), Queretaro, 2015. <http://www.qro.cinvestav.mx/%0A~aherrera/reportesInternos/unknownOri ginShirley.pdf>.

3:00pm **AS+CA+EL+EM+SE+SS+TF-WeA-3 Chemical Analysis of Multilayer System by Photoemission: The Binding Energy Reference Challenge**, Thierry Conard, A. Vanleenhove, IMEC Belgium; D. Desta, H. Boyen, University of Hasselt, Belgium

XPS is a well-established technique used for non-destructive analysis of the chemical composition of thin layers and interfaces. It is most commonly performed using Al K $\alpha$  radiation (1486.6 eV), which limits the analysis to the top 5-10nm. The recently developed laboratory-based hard X-ray photoelectron spectrometers (HAXPES) provide new analysis options. They enable the analysis of thicker film structures and interfaces buried down to 20-50 nm depending on the photon energy and facilitate the analysis of fragile buried layers without ion-induced chemical damage.

Increasing the number of analyzed (insulating) layers enhances the risk of significant vertical differential charging and makes the repeatability of binding energy determination more challenging. While charging has to be taken into account for XPS, the analysis of most XPS spectra is quite straightforward as long as the surface charge is stable and the lateral distribution of surface charge is uniform within the area of analysis. For HAXPES however vertical charge distribution comes into the game for a large group of structures whose development can benefit from HAXPES analysis. Vertical charge build-up can be complex, especially if examined structures exist of multiple layers and hence multiple interfaces, containing a large variety of materials. But even in 'simple' non-conducting one-layer structures a vertical charge gradient builds up when exposed to X-rays and small changes in the parameters of standard surface charge neutralization techniques - as the use of e-beam flood guns - can influence the nature of the charge gradient.

In this work, we will examine the influence of measurement conditions in single and multiple layers systems relevant to the microelectronic industry on the determination of precise binding energies.

HAXPES spectra of technologically relevant samples will be discussed to demonstrate the challenge of determining exact binding energy values. The set of examined samples comprises oxide layers Si samples and metal/high-k/Si stacks including high-k materials as HfO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub>. The surface potential will be precisely set and monitored in situ by depositing a metallic layer (such as Ag) on top of the stack and applying an external potential instead of using an electron neutralization source. All experiments are performed in a PHI Quantex system and/or a Scienta Omicron HAXPES Lab, both equipped with two monochromatic X-ray sources: an Al K $\alpha$  (1486.6 eV) and a Cr K $\alpha$  (5414.8 eV - Quantex) or Ga K $\alpha$  (9252.1 eV - HAXPES lab) X-ray source.

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3:20pm **AS+CA+EL+EM+SE+SS+TF-WeA-4 Where Are We on the Road-Map to Artificially Intelligent Interpretation of X-ray Photoelectron Spectra?**, C. Moffitt, Kratos Analytical Inc; A. Roberts, J. Counsell, C. Blomfield, Kevin Good, K. Macak, Kratos Analytical Limited, UK

Robust peak identification is crucial for accurate sample analysis using X-ray photoemission spectroscopy (XPS). Automation of peak ID enhances this process by minimizing user error and bias. Current acquisition software offers improved computer-derived peak identification from unknown samples, instilling confidence in the correct identification of elements. Moreover, this forms the foundation for an automated sample analysis workflow known as Data-dependent Analysis (DDA). DDA involves identifying peaks in a survey spectrum and subsequently acquiring high-resolution spectra from major components. A recent User survey revealed that a significant majority of users rely on the large area survey acquisition mode as a starting point for analysis.

To provide a metric for confidence in the DDA process, existing spectral analysis data, which includes the element composition information, is used to generate reference spectra for testing purposes. These reference spectra serve as the basis dataset against which the performance of the automated analysis algorithm can be evaluated. By comparing the results of the algorithm with the reference spectra, statistical parameters can be calculated to assess the algorithm's precision, sensitivity, specificity, and accuracy in identifying elements of unknown spectra.

For experienced analysts, DDA serves as a time-saving acquisition method, while for inexperienced analysts, it provides assurance in accurate peak identification and appropriate high-resolution spectra acquisition. Here we highlight current status of automated XPS data acquisition in relationship to the 'expert system', championed in the early 2000's and full AI interpretation of XPS spectra of the future.

4:20pm **AS+CA+EL+EM+SE+SS+TF-WeA-7 Thin Film Analysis by XPS: Quantitative Analyses Using Physics-Based and Machine-Learning Approaches**, Lev Gelb, N. Castanheira, A. Walker, University of Texas at Dallas

We present progress towards quantitative analysis of XPS data using both model-based "fitting" approaches and machine learning methods. Two separate applications are considered.

The first concerns the simultaneous extraction of both compositional profiles and sputtering parameters from XPS sputter depth-profiles of multilayer films. Depth-profile data are routinely processed to provide "fractional composition vs ion dose" profiles, but such analyses typically assume the sample is homogeneous in the probed region, which is not true near interfaces, and cannot precisely convert between units of ion dose and depth without extensive calibration data. Our approach is to first construct analytical models for both the sample structure and for the sputtering process, and then to determine the model parameters (layer thicknesses, interfacial widths, material removal rates, etc.) that are most likely given the observed apparent fractional composition profiles. This is done numerically, by iteratively comparing simulated and observed apparent composition profiles. The only additional required inputs are the inelastic mean free paths for each tracked peak in each material present. The efficacy of this approach is demonstrated using both synthetic and experimental data sets, and various model improvements (sputter-induced mixing, *in situ* chemical reactions) are discussed.

The second application concerns the application of machine-learning tools to remove the inelastic scattering background from XPS spectra in order that accurate peak areas can be obtained. Our approach here is to generate a training data set which consists of a thousands of simulated XPS spectra with and without inelastic scattering included. This is accomplished using the SESSA software package[1]. This data set is then used to train a neural network algorithm to output a "no-background" spectrum from an input "with-background" spectrum; this output spectrum can then be used to compute peak areas for compositional analysis. The training set generation methodology and network structure are discussed, and application of the tool to both simulated and experimental spectra is demonstrated.

[1] Werner, W., Smekal, W., Powell, C. and Gorham, J. (2021), *Simulation of Electron Spectra for Surface Analysis (SESSA) Version 2.2 User's Guide*, Natl Std. Ref. Data Series (NIST NSRDS), <https://doi.org/10.6028/NIST.NSRDS.100-2021>.

4:40pm **AS+CA+EL+EM+SE+SS+TF-WeA-8 Room Temperature Ionic Liquids as Reference Materials for Photoelectron Spectrometers**, Benjamin Reed, National Physical Laboratory, U.K.; J. Radnik, BAM Berlin, Germany, UK; A. Shard, National Physical Laboratory, U.K.

Room-temperature ionic liquids (RTILs) are materials consisting of organic salts that are liquid below temperatures of 100°C and are used in several fields including electrochemistry,<sup>1</sup> pharmaceuticals, and medicine.<sup>2</sup> RTILs have several notable properties that make them ideal for X-ray photoelectron spectroscopy (XPS) analysis. They have an extremely low vapor pressure and high surface tension, and so can be analysed using conventional XPS under ultrahigh vacuum without the need for near-ambient pressure instrumentation. Also, when deposited in a recessed sample holder, the meniscus of an RTIL will be perfectly flat meaning that there are no contributions from sample topographic effects. Finally, and most importantly, they are highly homogeneous and have well-defined stoichiometries.<sup>3</sup>

These properties make RTILs potential reference materials for validating the intensity calibration of a photoelectron spectrometer. RTILs with non-coordinating bistriflimide (NTf<sub>2</sub>) anions (e.g. PMIM<sup>+</sup>NTf<sub>2</sub><sup>-</sup>) or dimethyl phosphate (DMP) anions (e.g. MMIM<sup>+</sup>DMP<sup>-</sup>) are such candidates, with core levels up to ~800 eV binding energy, making them apt for verifying the quantification of light elements, especially for organic materials.<sup>4,5</sup>

To accurately determine peak areas, however, requires the principal and secondary photoelectron signals to be deconvolved. Previous attempts by multiple laboratories using different quantification methods give a mean atomic composition within 1 at.% of the known stoichiometry, but some individual elements (such as fluorine) exhibit differences greater than 1 at.% because the elastic and inelastic secondaries are not suitably deconvolved. Attention must be paid to the energy loss function that defines the inelastic background over the full energy range of an XPS spectrum so that a suitable Tougaard background subtraction can be applied.<sup>6</sup> Here we present a study on several RTILs and discuss how they may be used to validate an XPS intensity calibration and provide confidence in measurements to XPS instrument operators.

<sup>1</sup>M. Armand, F. Endres, D. R. MacFarlane et al., *Nat. Mater.* **8**, 621 (2009).

<sup>2</sup>K. S. Egorova, E. G. Gordeev, and V. P. Ananikov, *Chem. Rev.* **117**, 7132 (2017).

<sup>3</sup>E. F. Smith, I. J. Villar Garcia, D. Briggs et al., *Chem. Commun.* **45**, 5633 (2005).

<sup>4</sup>B.P. Reed, J. Radnik, and A.G. Shard, *Surf. Sci. Spectra* **29**, 014001 (2022).

<sup>5</sup>X. Knigge and J. Radnik, *Surf. Sci. Spectra* **30**, 014006 (2023).

<sup>6</sup>M. P. Seah, I. S. Gilmore, and S. J. Spencer, *Surf. Sci.* **461**, 1 (2000).

5:00pm **AS+CA+EL+EM+SE+SS+TF-WeA-9 Fractional Coverage Analysis of Monolayers with XPS and Non-Destructive Depth-Profiling with Combined Soft and Hard X-Rays**, Norbert Biderman, K. Artyushkova, D. Watson, Physical Electronics USA

X-ray photoelectron spectroscopy (XPS) is a well-established technique for non-destructive analysis of the chemical composition of thin layers and interfaces. Angle-resolved XPS (AR-XPS) has been used to determine composition of depth profiles and layer thicknesses, traditionally with Al K $\alpha$  (1486.6 eV) X-ray beams for depths up to 5-10 nm below the surface. In recent years, new AR-XPS capabilities have been added to Physical Electronics XPS scanning microprobe instruments including Cr K $\alpha$  (5414.8 eV) hard X-ray photoelectron spectroscopy (HAXPES) that can probe buried interfaces up to 15-30 nm below the surface. Coinciding with the HAXPES development, the StrataPHI analysis software was developed to reconstruct quantitative, non-destructive XPS/HAXPES depth profiles from angle-dependent and single-angle photoelectron spectra.

In this talk, we will show that the StrataPHI software has been further developed to combine Al K $\alpha$  and Cr K $\alpha$  XPS and HAXPES data within a single depth profile to enhance extracted analytical information from various depths below the surface. We will explore the method of the combined technique as well as its application to multilayered thin film samples. The updated StrataPHI software also includes a fractional coverage analysis mode, relevant in situations where ultra-thin films exist as discrete islands – commonly observed in early thin-film deposition stages on the substrate rather than as a continuous, uniform film. A model system of discrete molybdenum sulfide (MoS<sub>2</sub>) monolayer triangles deposited on SiO<sub>2</sub>/Si substrate will be discussed.

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Such added StrataPHI capabilities allow for scientists and engineers in metrology and research & development to analyze multi-layered thin films and ultra-thin films rapidly and non-destructively without potentially damaging ion beam sputtering that might otherwise be required to depth-profile or sputter-clean adventitious contamination off the surface.

5:20pm **AS+CA+EL+EM+SE+SS+TF-WeA-10 Reassessing the Reduction of Ceria in X-Ray Photoelectron Spectroscopy**, *David Morgan*, Cardiff University, UK

Given its excellent redox abilities, the use of cerium dioxide ( $\text{CeO}_2$ , ceria) and related materials in catalysis is widespread [1]. This  $\text{Ce}^{3+}/\text{Ce}^{4+}$  redox shuffle allows for great catalytic ability and a method of correlation of catalytic activity to the state of ceria [2–4]. Given that catalysis is a surface mediated process, XPS is critical in the analysis of pre- and post-mortem materials.

Over the years there has been debate on the degree of reduction of  $\text{CeO}_2$  during XPS analysis. Therefore, in continuation of our work on understanding the reduction of materials in modern spectrometers [5], we have investigated different cerium oxide preparations and shown that not only is the rate of reduction dependent on instrument type and experimental configuration (and hence appropriate analysis protocols should be implemented), but is also related to the morphology of the cerium which may, at least in part, account for the discrepancies in the degree of reduction in the literature. It is postulated that reduction rates could be used to indicate likely ceria morphology where other analysis is unavailable.

## References

[1] Catalysis By Ceria And Related Materials, 2nd Edition.; Trovarelli, A., Fornasiero, P., Eds.; Imperial College Press: London, 2013.

[2] Smith, L. R.; Sainna, M. A.; Douthwaite, M.; Davies, T. E.; Dummer, N. F.; Willock, D. J.; Knight, D. W.; Catlow, C. R. A.; Taylor, S. H.; Hutchings, G. J. "Gas Phase Glycerol Valorization over Ceria Nanostructures with Well-Defined Morphologies". *ACS Catal*, 2021, 11 (8), 4893–4907.

[3] Qiao, Z.-A.; Wu, Z.; Dai, S. "Shape-Controlled Ceria-Based Nanostructures for Catalysis Applications". *ChemSusChem*, 2013, 6 (10), 1821–1833.

[4] Ziemba, M.; Schilling, C.; Ganduglia-Pirovano, M. V.; Hess, C. "Toward an Atomic-Level Understanding of Ceria-Based Catalysts: When Experiment and Theory Go Hand in Hand". *Acc Chem Res*, 2021, 54 (13), 2884–2893.

[5] Morgan, D. J. "XPS Insights: Sample Degradation in X-ray Photoelectron Spectroscopy". *Surface and Interface Analysis*, 2023. (In Press)

## Acknowledgements

This work acknowledges the EPSRC National Facility for XPS ('HarwellXPS'), operated by Cardiff University and UCL, under contract No. PR16195, and C.M.A. Parlett and X. Zhou for provision of nanostructured ceria materials.

5:40pm **AS+CA+EL+EM+SE+SS+TF-WeA-11 Using High Sensitivity – Low Energy Ion Scattering Spectroscopy (LEIS) to Unravel the Complex Nature of High Entropy Alloys**, *Matthias Kogler, C. Pichler*, Centre for Electrochemistry and Surface Technology (CEST GmbH), Austria; *M. Valtiner*, Vienna University of Technology, Austria

Complex metallic materials such as Multi-Principal Alloys (MPEAs) and High Entropy Alloys (HEAs) have emerged as a promising class of materials given their unique inherent characteristics. Excellent mechanical, thermal, and corrosion properties allow for a broad spectrum of applications. However, due to the multi-element nature of these alloys, characterisation of the composition and microstructure proves to be a challenging task.

Especially with regard to corrosion-protective passivation films, the complex correlations with the corrosion behaviour are fully unclear to date, and require an in-depth atomic level characterisation and rationalisation. However, the precise layer by layer structure of such passive films is particularly demanding to assess, since traditional techniques such as XPS (X-ray photoelectron spectroscopy) or AES (Auger electron spectroscopy) have analysis penetration depths of several nanometres and cannot reach

atomic layer resolution. However, to fully understand and quantify the passivation layer structure, such an atomic layer resolution of the surface region is necessary, due to the complexity of HEAs.

In order to obtain an exact understanding of the atomistic mechanism at the monoatomic layer level, High-Sensitivity - Low Energy Ion Scattering Spectroscopy (HS-LEIS), was applied, which provides the required monolayer sensitive resolution to study the passivation layers of such complex multi-component alloys. The unique surface sensitivity combined with the implementation of in-situ treatment methods enabled the real-time study of oxide layer growth, as well as the analysis of temperature-dependent changes in the elemental surface composition. Due to the high resolution achieved by static and dynamic sputter depth profile modes, we could determine the exact composition of the HEA passivation layer with resolution on atomic monolayer scale.

The findings provide the potential to significantly advance the current understanding of the passivation behaviour of MPEAs and HEAs, and the development of novel metallic materials with superior properties. Valuable insights for understanding the material characteristics for those highly advanced materials could thereby be generated.

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